

# Wurtzite-Perovskite-Wurtzite (ZnO-BaTiO<sub>3</sub>-ZnO) Interface Polarization Hysteresis Model



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## Multifunctional 3D Heterostructures



D. E. Scharrett and R. E. Garrison *Ann. Log.* 37 (2005).

Multi functional capabilities of **Future electronics.**

New concepts incorporate ferroelectric and piezoelectric properties into semiconductors to make multifunctional architectures for future device technology.

Here we introduce a multilayered structure prepared with wurtzite ZnO and perovskite BaTiO<sub>3</sub>.

The coupling between the ZnO polarization (surface ionic polarization charge  $P_{sz}$ ) and the switchable ferroelectric perovskite BaTiO<sub>3</sub> polarization  $P_d$  influences:

(I) Ferroelectric refractive index change  $\Delta n$  [1].

(II) Ferroelectric phase transition [2, 3].

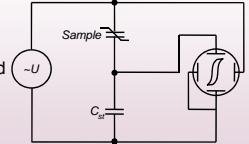
(III) Electrical properties [4-6].

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## Experiment

ZnO, BaTiO<sub>3</sub> heterostructures are prepared by Pulsed Laser Deposition and subsequent masking with ohmic Pt back and front contacts.

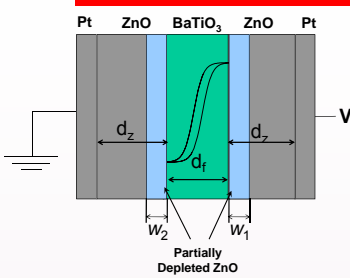
Electric Sawyer-Tower [5] measurements were performed on samples at 1.5 kHz frequency respectively.



Sawyer-Tower circuit

## Electric Interface Polarization Coupling and Depletion Layer Model

### Internal mechanism



Voltage distribution across the structure:

$$V = d_f E_f + \frac{e N_c}{2 \epsilon_z} (w_1^2 - w_2^2).$$

Internal change mechanism:

$$V = \frac{d_f}{\epsilon_f} \sigma_b - \frac{d_f}{\epsilon_f} P_d - \frac{e N_c}{2 \epsilon_z} (w_2^2 - w_1^2) - \frac{d_z}{\epsilon_z} (P_{sz1} + P_{sz2}).$$

Boundary conditions at right and left ZnO-BaTiO<sub>3</sub> interfaces:

$$e N_c w_1 + P_{sz1} = E_f \epsilon_f + P_d,$$

$$-e N_c w_2 + P_{sz2} = E_f \epsilon_f + P_d.$$

### Different model cases and the dipole polarization

Depending on the applied bias four different model cases are possible:

Case I : Both depletion widths,  $w_1$  and  $w_2$  are zero.

Case II : Depletion width  $w_1$  is zero but not  $w_2$ .

Case III : Depletion width  $w_2$  is zero but not  $w_1$ .

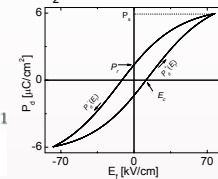
Case IV : Both depletion widths  $w_1$  and  $w_2$  are non-zero.

Dipole polarization ( $P_d$ ):

$$P_d^+(E_f) = P_s \tanh \left[ \frac{(E_f - E_c)}{2\delta} \right]$$

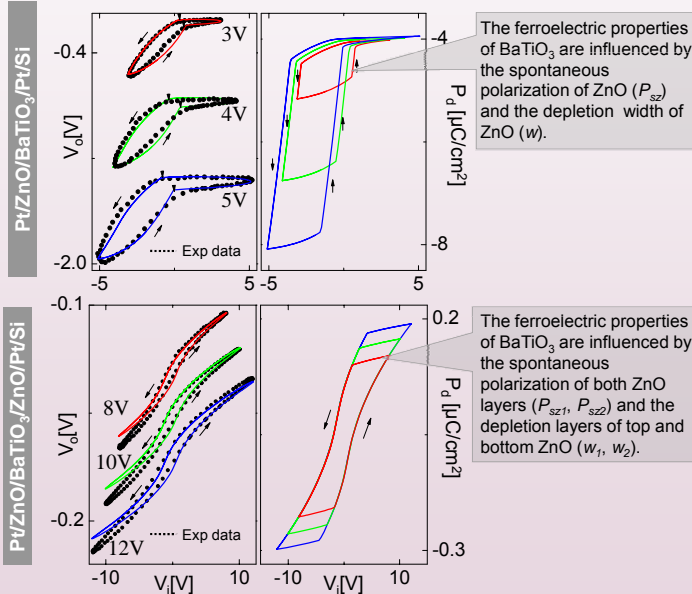
$$P_d^-(E_f) = -P_d^+(-E_f)$$

$$\text{where } \delta = E_c \left[ \log \left( \frac{1 + P_r/P_s}{1 - P_r/P_s} \right) \right]^{-1}$$



## Experimental and Model Calculated Data

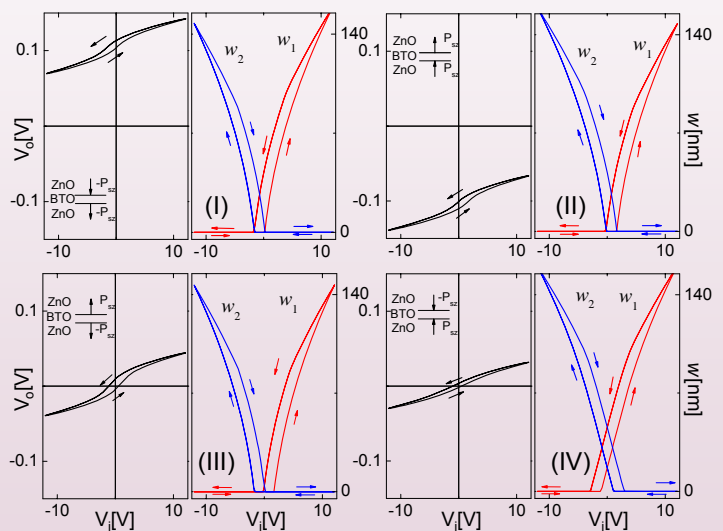
### Best match model results



The ferroelectric properties of BaTiO<sub>3</sub> are influenced by the spontaneous polarization of ZnO ( $P_{sz}$ ) and the depletion width of ZnO ( $w$ ).

The ferroelectric properties of BaTiO<sub>3</sub> are influenced by the spontaneous polarization of both ZnO layers ( $P_{sz1}$ ,  $P_{sz2}$ ) and the depletion layers of top and bottom ZnO ( $w_1$ ,  $w_2$ ).

### Effect of ZnO polarity orientation in ZnO-BaTiO<sub>3</sub>-ZnO



>Orientation of spontaneous polarization of the ZnO layers controls the ferroelectric properties of the ZnO-BTO-ZnO structure.

>Either positive or negative orientations of both spontaneous polarizations of ZnO layers results an asymmetry in the transport properties (different depletion layer endings in fig. (I) and (II)).

>Asymmetric transport properties of the structure results in an asymmetric shift in the Sawyer-Tower response (shown in fig(I) and (II)).

Model parameters		ZnO/BaTiO <sub>3</sub>	ZnO/BaTiO <sub>3</sub> /ZnO
Coercive field ( $E_c$ )	V/m	1.1	1
Spontaneous polarization of BaTiO <sub>3</sub> ( $P_s$ )	$\mu\text{C}/\text{cm}^2$	14.1	2
Remanent polarization of BaTiO <sub>3</sub> ( $P_r$ )	$\mu\text{C}/\text{cm}^2$	5.64	0.06
Spontaneous polarization of ZnO ( $P_{sz}$ )	$\mu\text{C}/\text{cm}^2$	-4.1	0.1
Overall sample resistance ( $R_s$ )	k $\Omega$	13	420
Dielectric constant of BaTiO <sub>3</sub> ( $\epsilon_f$ )		250	60
Dielectric constant of ZnO ( $\epsilon_{ZnO}$ )		8	8

[1] V. Voora et al., *phys. stat. sol. (c)* 5, 1328 (2008).

[4] V. Voora et al., *J. Electron. Mater.* 37, 1029 (2008).

[2] M. Schubert et al., *Ann. Phys.* 13, 61 (2004).

[5] N. Ashkenov et al., *Thin Solid Films* 486, 153 (2005).

[3] B. Mbenkum et al., *Appl. Phys. Lett.* 86, 091904 (2005).

[6] V. Voora et al., *Appl. Phys. Lett.* 94, 142904 (2009).